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Growth, single-crystal structure, and magnetic properties of the double molybdate $\text{KYb}(\text{MoO}_4)_2$

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Abstract

Crystals of the rare earth double molybdate $\text{KYb}(\text{MoO}_4)_2$ were grown by spontaneous nucleation from a $\text{K}_2\text{Mo}_4\text{O}_{13}$ flux. The typical crystals were colorless and elongated, well-developed plates. The material has been characterized by powder X-ray diffraction (PXRD), energy-dispersive X-ray spectrometry (EDS), and Raman spectroscopy methods. The structural properties have been investigated through single-crystal X-ray diffraction at room temperature. $\text{KYb}(\text{MoO}_4)_2$ crystallizes in the orthorhombic symmetry, space group $Pbcn$ (No. 60), $a = 5.0395(2)$, $b = 18.2821(9)$, $c = 7.8635(4)$ Å, and $Z = 4$. A complete ordering of the Yb^{3+} and Mo^{6+} cations has been found. The Yb^{3+} ions are located in distorted YbO_8 antiprisms which form chains of edge-sharing, aligned along the c -axis while the Mo^{6+} ions are placed in a slightly deformed tetrahedral environment of oxygen atoms. Double bridge modes of the " Mo_2O_8 dimers" have been identified in the unpolarized Raman spectra. The temperature-dependent magnetic susceptibility of $\text{KYb}(\text{MoO}_4)_2$ follows a Curie–Weiss law with a Weiss constant of -52 K.

Keywords: A. oxides; B. crystal growth; C. X-ray diffraction; D. crystal structure; D. magnetic properties

1. Introduction

Double molybdates and tungstates (DMo / DW) of general formula $AT(XO_4)_2$ with $X = Mo^{VI}$ or W^{VI} , a monovalent element A and a trivalent or lanthanide element T, exist with a wide variety of polymorphic forms, mainly tetragonal and orthorhombic, which largely depend on the ratio of the ionic radii A^+/T^{3+} [1-6]. This structural diversity results in several properties such as negative thermal expansion, third-harmonic generation or ionic conduction, and applications in modern technology [7-11]. Special attention has been paid to trigonal DMo and DW due to their simple structure and their ability to undergo ferroelastic phase transitions over wide temperature ranges [2, 11]. Some DMo and DW phases, such as orthorhombic ($Pbna$) $KYMo_2O_8$, have potential applications as microchip lasers due to their perfect cleavage in the (100) planes [2, 8, 10].

The compounds $AT(XO_4)_2$ have generated some interest as new materials for laser systems because the isostructural substitution of the trivalent element T by optically active lanthanides Ln is possible over a wide concentration range, sometimes up to the stoichiometric composition. For example, $KGd(WO_4)_2$ is well-known as an efficient Raman material that contains an ionic molecular complex that exhibits active oscillations of Stimulated Raman Scattering [12].

Moreover, the lanthanide-based, magnetically concentrated crystals of double tungstates and molybdates are shown to display optical properties such as photon upconversion emission as well as interesting magnetic properties [13, 14]. Single-molecule magnets (SMM) and single-ion magnets (SIM) systems based on oxygen coordination polyhedra around Nd^{3+} and/or Yb^{3+} lanthanides have been synthesized [15, 16].

The present work is part of a study aimed at the synthesis and structural evaluation of the Ln-based magnetically-concentrated double molybdates, $KLn(MoO_4)_2$. The formation of several

molybdates of this family has already been reported in the past [1-3, 8, 10, 15, 17-21] but very fragmentary information is available for $\text{KYb}(\text{MoO}_4)_2$ [14, 19, 21].

We present here the flux-growth of the orthorhombic phase $\text{KYb}(\text{MoO}_4)_2$ and its single-crystal structure determined at room temperature. The magnetization and hysteresis loops have been measured in a temperature range from 1.8 to 300 K to study its magnetic behavior.

2. Experimental

2.1. Material preparation

The starting reagents were commercial Alfa Aesar products MoO_3 (99.95%), K_2CO_3 (99%), and Yb_2O_3 (99.9%), and were used as received. Crystals were obtained using the high-temperature flux growth technique through spontaneous crystallization. The $\text{K}_2\text{Mo}_4\text{O}_{13}$ flux was previously synthesized *via* a solid-state reaction at 500 °C for two weeks, from powders of K_2CO_3 alkali carbonate (31 g) and MoO_3 molybdenum oxide (129 g). A total amount of 20 g of an intimate mixture of Yb_2O_3 and $\text{K}_2\text{Mo}_4\text{O}_{13}$ in a weight ratio of 10/90 was then placed in a 5 cm³ Pt crucible and heated in an electrical furnace at 970 °C in air for 6 hours to ensure the complete dissolution of the solute. Then, the temperature was slowly lowered at a rate of 1°/h down to 700 °C for crystal growth, then the furnace was turned off. Crystals were obtained at the top of the solidified flux from which they were separated by dissolving the flux in water.

2.2. Material characterization

The crystallized material was characterized by classical EDX analyses, Raman spectroscopy, and X-ray diffraction. More details on the experimental conditions can be found in a previous paper [22]. Le Bail profile refinement has been performed using the program Jana2006 [23].

A stereo microscope with a polarizing filter was used to select a single crystal for X-ray diffraction data collection. Diffracted intensities were measured at room temperature on an Xcalibur CCD diffractometer (Oxford diffraction, Mo $K\alpha$ radiation) and corrected for Lorentz-polarization effects using CrysAlis software [24]. Absorption correction was applied using the included numerical procedure. The structure was solved and refined using the SHELX programs [25, 26] and anisotropic displacement parameters were considered for all atoms. For all atoms, the site occupation factors were freely refined and remain close to their normal values (within 2σ limits) indicating the complete filling of the sites and a good agreement of the refined stoichiometry with the analysis of crystals.

Magnetization (M) was measured using a commercial Superconducting Quantum Interference Device (SQUID) MPMS-7XL magnetometer in DC mode. Temperature-dependent magnetization was measured between 1.8 and 300 K in an intermediate H field of 5 kOe (0.5 Tesla). Magnetic susceptibility χ was defined as M/H. The field-dependent magnetization was measured between 1.8 and 300 K using the same equipment with magnetic fields up to 5 Tesla.

3. Results and discussion

3.1. Flux and crystals characterizations

The formation of the $K_2Mo_4O_{13}$ flux in the powder form was characterized by room-temperature X-ray powder diffraction, Figure 1. It was positively compared with the Powder Diffraction Standards Card N° 00-27-0416 [27].

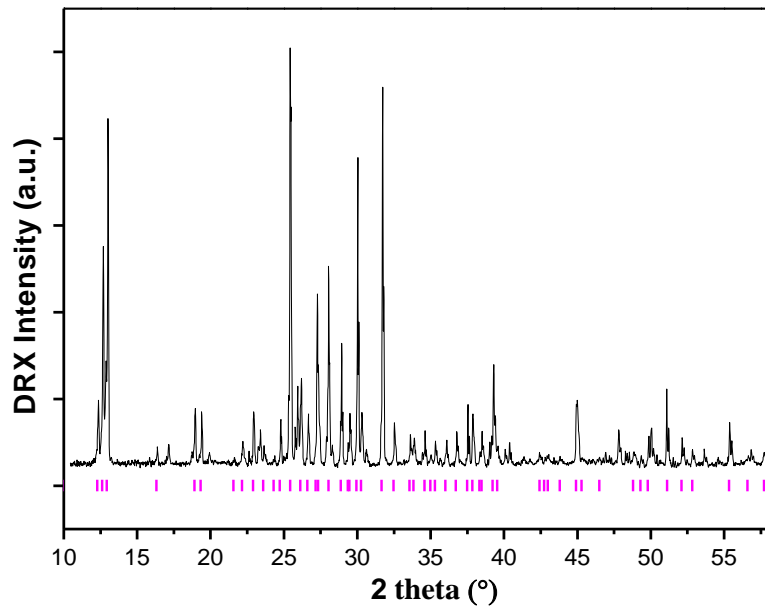


Figure 1. XRD experimental powder pattern ($\lambda = 1.5418 \text{ \AA}$) of $\text{K}_2\text{Mo}_4\text{O}_{13}$ indexed with the Bragg positions (pink bars) of the $\text{K}_2\text{Mo}_4\text{O}_{13}$ PDF card [27].

Spontaneous nucleation yielded sub-millimeter and millimeter-sized plate-shaped crystals, and no post-growth annealing was applied. Figure 2 shows selected as-grown crystals with dimensions up to $6.0 \times 1.2 \times 0.5 \text{ mm}$. The transparent and colorless crystals display good optical quality, they are free of macroscopic inclusions and present sharp edges.

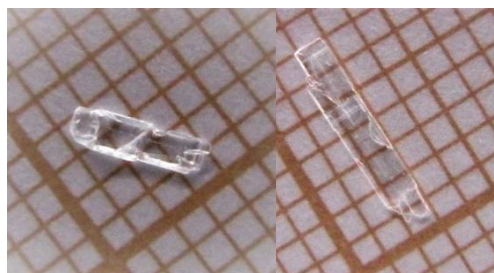


Figure 2. Selected as-grown crystals of $\text{KYb}(\text{MoO}_4)_2$ shown on millimeter grid paper.

The chemical composition was checked by EDX analysis on an as-grown single crystal (not polished sample), and no significant variation in stoichiometry was observed when scanning across the crystal. The elements K, Yb, Mo, and O were found in a molar ratio 7.0(3):10.34(6):17.5(4):65.1(7) which leads to the compound formula $\text{K}_{0.80(5)}\text{Yb}_{1.18(2)}\text{Mo}_{2.00(2)}\text{O}_{7.44(3)}$.

The Cu $K\alpha$ XRD powder pattern registered for a crushed sample in standard conditions at room temperature is given in Figure 3. All the diffraction peaks in the 2θ range from 5 to 70° were indexed in agreement with the JCPDS card n° 00-050-1762 [28] which announces $\text{KYb}(\text{MoO}_4)_2$ to crystallize in the orthorhombic space group D_{2h}^{14} (*Pbcn*) with four formula units per unit cell.

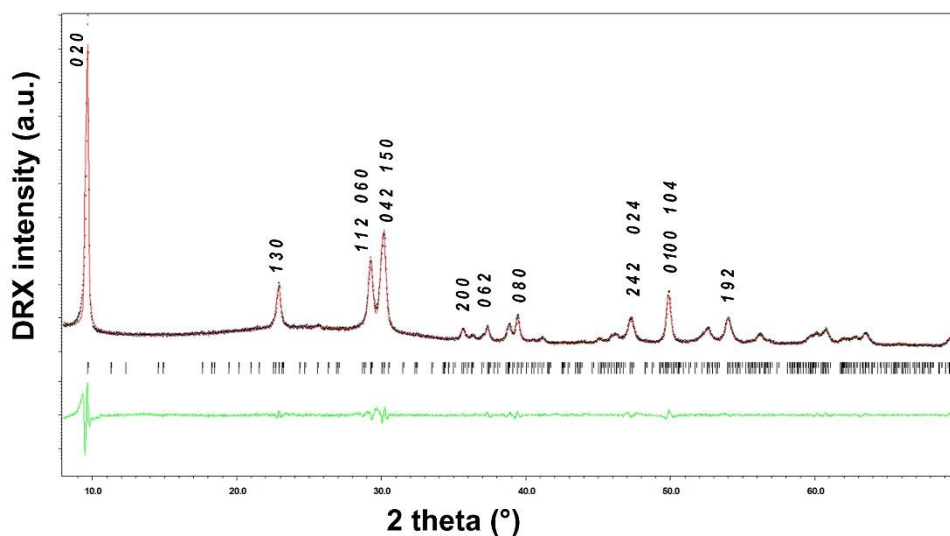


Figure 3. XRD experimental (black data points) powder pattern ($\lambda = 1.5418 \text{ \AA}$) of $\text{KYb}(\text{MoO}_4)_2$, calculated profile (red line) and difference curve (green line). Cell parameters converged to $a = 5.045(2)$, $b = 18.297(4)$, $c = 7.852(2) \text{ \AA}$ in Le Bail refinement.

These results confirm the formation, under the stated experimental conditions, of the crystallized stoichiometric phase $\text{KYb}(\text{MoO}_4)_2$ containing the magnetic ytterbium element in its stable Yb^{3+} oxidation state.

3.2. Single-crystal structure

The crystal displayed orthorhombic symmetry and the reflections were indexed in a lattice with $a = 5.0395(2)$, $b = 18.2821(9)$, $c = 7.8635(4)$ Å. Since the statistics indicate centrosymmetric and reflections $0kl: k = 2n + 1$, $h0l: l = 2n + 1$, $hk0: k + l = 2n + 1$ were systematically absent from the data set, the space group *Pbcn* (No. 60) was selected. The structural model found from direct methods includes one Mo and four oxygens at general $8d$ positions as well as Yb and K at special $4c$ positions with C_2 local symmetry. This leads to four $KYb(MoO_4)_2$ units in the orthorhombic cell and leads to assign the $KY(MoO_4)_2$ structural type [17] to the compound. During the refinement, discrepancies are noted for some reflections, seemingly poorly measured. These problems result from the existence of a parasitic crystallite hanging on the main crystal that affects the measure of intensities in the corresponding domain and consequently the agreement factors values. However, the final structural refinement was carried out to an R_1 agreement factor of 5.57 %. The main crystal data and structure refinement details are reported in Table 1, while the atomic positions and equivalent displacement parameters are listed in Table 2. The full CIF file with the CSD number 2049313 can be freely obtained from the Cambridge Crystallographic Data Center (CCDC) [29].

Table 1. Crystal data, structure refinement, and data collection details

Crystal data	
Formula, Z	KYb(MoO ₄) ₂ , 4
Symmetry, space group	Orthorhombic, <i>Pbcn</i> (60)
Lattice dimensions (Å)	<i>a</i> = 5.03950(19), <i>b</i> = 18.2821(9), <i>c</i> = 7.8634(4)
Lattice volume (Å ³)	<i>V</i> = 724.48(6)
Crystal dimensions (mm ³)	0.29 × 0.16 × 0.04
Calculated density	4.878
Data collection	
Temperature (K)	298
Radiation, wavelength (Å)	Mo Kα, 0.71073
θ range (°)	3.418 - 32.578
Limiting indices	-7 ≤ <i>h</i> ≤ 7; -26 ≤ <i>k</i> ≤ 27; -11 ≤ <i>l</i> ≤ 11
Scan mode	ω scans
Absorption correction; T _{min} , T _{max}	numerical, 0.10, 0.62
Absorption coefficient (mm ⁻¹)	16.79
Refinement	
Method	Full-matrix least-squares on F ²
Collected / independent reflections	12880 / 1283 [R _{int} = 0.0340]
Data / restraints / parameters	1283 / 0 / 56
Final R indices [I > 2σ(I)]	<i>RI</i> = 0.0557, <i>wR2</i> = 0.1163
R indices (all data)	<i>RI</i> = 0.0592, <i>wR2</i> = 0.1173
Density residuals (e.Å ⁻³)	3.99 and -3.58

Older reports [18, 28, 30] gave, for the magnetically concentrated KYb(MoO₄)₂ compound, the orthorhombic cell parameters. A structural type was assigned from powder patterns analogy with KY(MoO₄)₂ structure [17] but no structural refinement is available in the literature. The present study brings the first complete room-temperature structure determination of KYb(MoO₄)₂ double molybdate from single-crystal data.

Table 2. Atom coordinates ($\times 10^4$) and equivalent displacement parameters ($\text{\AA}^2 \times 10^3$) for $\text{KYb}(\text{MoO}_4)_2$. U_{eq} is defined as one-third of the trace of the orthogonalized U_{ij} tensor.

	Wyckoff position	x	y	z	U_{eq}
Yb	4c	0	49(1)	2500	10(1)
K	4c	0	-2276(2)	2500	22(1)
Mo	8d	4776(2)	1003(1)	-144(1)	10(1)
O(1)	8d	7516(16)	337(5)	67(11)	12(1)
O(2)	8d	6043(19)	1877(5)	-198(13)	18(2)
O(3)	8d	2574(17)	926(5)	-1912(11)	13(2)
O(4)	8d	2658(19)	987(5)	1645(10)	15(2)

The crystal structure of $\text{KYb}(\text{MoO}_4)_2$ drawn with the program Diamond 3.2 [31] is presented in Figure 4, and the main interatomic distances and angles are gathered in Table 3. Looking along the b -axis, this structure can be viewed as a multilayered structure with a sequence of YbMo_2O_8 complex layers alternated with wavy layers of potassium, which provides the structure with a certain two-dimensional character. The potassium present in the structure as K^+ cations acts as a link between the YbMo_2O_8 layers through bonding ionic interactions. Indeed, the K^+ cation has a coordination number of 6 with oxygen placed at distances slightly shorter than 2.828 \AA , the average distance observed for K^+ ions bonded to oxygen according to an extensive study of bond-length distributions [32]. The oxygen coordination polyhedron around K^+ ($2 \times 2.68(1)$, $2 \times 2.79(1)$, and $2 \times 2.82(1)$ \AA) is a highly distorted octahedron that displays the C_2 symmetry.

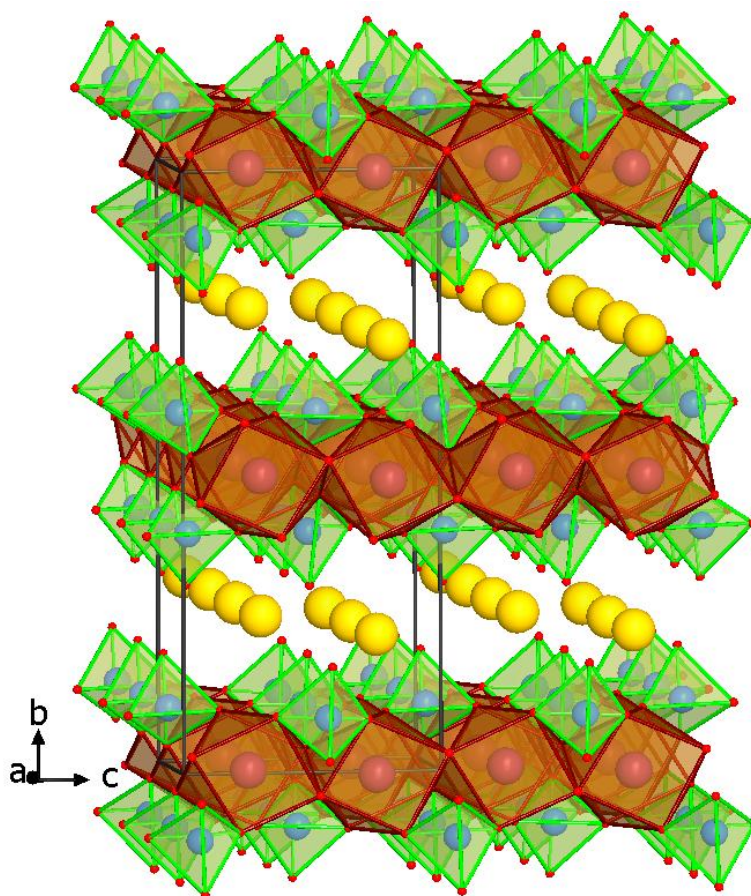


Figure 4. The orthorhombic $Pbcn$ structure of $KYb(MoO_4)_2$. Complex layers of MoO_4 (green) and YbO_8 (brown) polyhedra alternate with K^+ (yellow) layers

The $YbMo_2O_8$ complex layer can be viewed with a central sheet of YbO_8 polyhedra between two sheets of MoO_4 polyhedra (Figure 5). The Yb^{3+} ion sits inside a square antiprism of oxygen at distances from 2.253(9) to 2.478(8) Å, Table 3, which is characterized by only a two-fold axis along the b direction. Representation of the $YbMo_2O_8$ complex layer projected on the ac plane (Figure 4) shows that YbO_8 square antiprisms share one edge with each of their two alike neighboring units to form chains aligned along the c -axis. These YbO_8 chains are interconnected through MoO_4 distorted tetrahedral units that display the C_2 (almost C_{3v}) symmetry. The MoO_4 polyhedra are separated from each other since they have no common O, but they share their four O vertices with four neighboring YbO_8 units. With Mo-O distances from 1.722(9) to 1.848(8) Å and O-Mo-O angles between 104.1(4) and 118.9(4)°, Table 3, the

MoO₄ polyhedra are characterized by strong covalent bonding. Thus, the atoms are covalently bonded within the YbMo₂O₈ layers which interact with the potassium layers by ionic bonding in a three-dimensional network. The links between the Mo atoms of successive "layers" occur either through Mo-O-Yb-O-Mo or Mo-O-K-O-Mo paths.

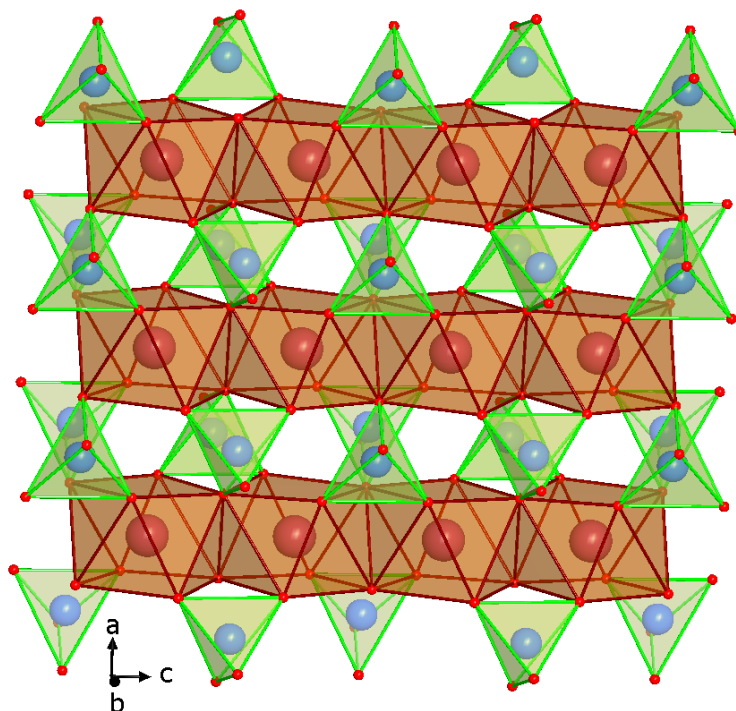


Figure 5. The YbMo₂O₈ complex layer (*b*-projection) built with YbO₈ chains (brown) aligned along the *c*-axis and interconnected through MoO₄ units (green).

It is interesting to focus on the connection mode of the YbO₈ polyhedra and notice the special arrangement of the MoO₄ units (Figure 6). The successive Mo sites along the *c*-axis are distant from either 3.712(1) Å or 4.164(1) Å bringing certain O atoms closer (or further away) from the Mo center. This results in the formation of Mo₂O₈ "dimeric-like" units involving two Mo-O(3) interactions of 2.676(8) Å (dashed orange lines, Figure 6) between the nearest MoO₄ units. Longer corresponding Mo-O(4) distances of 3.040(8) Å (dotted blue lines, Figure 6) are

found between the dimers. Similar "Mo₂O₈ dimers" have been described in the isostructural yttrium double molybdate [33, 34].

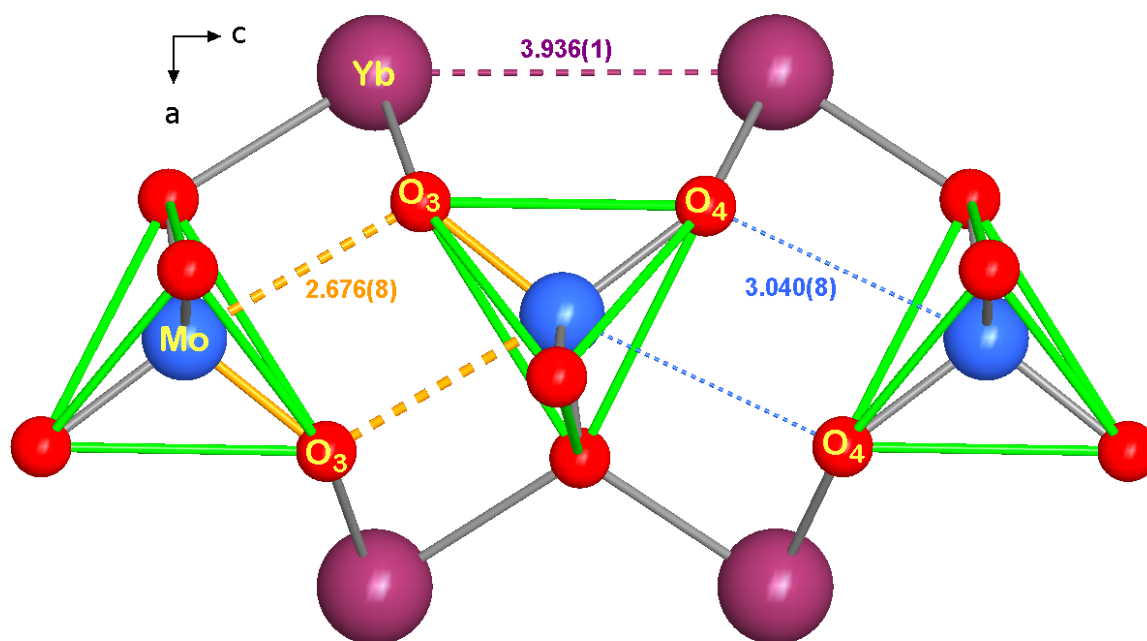


Figure 6. Organization of the MoO₄ pseudo-tetrahedral units showing double bridge Mo-O interactions of 2.676(8) Å (dash orange) within the "Mo₂O₈ dimer" and of 3.040(8) Å (dot blue) between the dimers.

The K⁺ and Yb³⁺ cations are fully ordered in the orthorhombic crystal KYb(MoO₄)₂ with the KY(MoO₄)₂ structural-type [17]. KYb(MoO₄)₂ double molybdate is isostructural to the KLn(MoO₄)₂ phases with Ln=Y, Dy, Ho, Er, Tm, Lu [17, 19, 30]. Instead, the double tungstate KYb(WO₄)₂, also with an ordered cation distribution, crystallizes in the monoclinic space group C2/c (C_{2h}⁶) (Z = 4) [35]. Besides, a monoclinic C2/c lattice is also given for the high-pressure form of the KYb(MoO₄)₂ compound which was assigned the structural type KYW₂O₈ based on its powder diffraction data [36].

Table 3. Selected geometric parameters in KYb(MoO₄)₂

<u>Distances to Yb</u>		<u>Angles around Yb</u>		<u>Angles around Mo</u>	
		O(3) ^{#1} -Yb-O(3) ^{#2}	75.3(4)		
Yb-O(3) ^{#1}	2.253(9)	O(3) ^{#1} -Yb-O(4)	150.9(3)		
Yb-O(3) ^{#2}	2.253(8)	O(3) ^{#2} -Yb-O(4)	108.5(3)	O(2)-Mo-O(4)	105.0(4)
Yb-O(4)	2.278(9)	O(3) ^{#1} -Yb-O(4) ^{#3}	108.5(3)	O(2)-Mo-O(3)	106.5(4)
Yb-O(4) ^{#3}	2.278(9)	O(3) ^{#2} -Yb-O(4) ^{#3}	150.9(3)	O(4)-Mo-O(3)	104.1(4)
Yb-O(1) ^{#4}	2.346(8)	O(4)-Yb-O(4) ^{#3}	82.3(5)	O(2)-Mo-O(1)	109.7(4)
Yb-O(1) ^{#5}	2.346(8)	O(3) ^{#1} -Yb-O(1) ^{#4}	130.7(3)	O(4)-Mo-O(1)	111.7(4)
Yb-O(1) ^{#6}	2.478(8)	O(3) ^{#2} -Yb-O(1) ^{#4}	72.7(3)	O(3)-Mo-O(1)	118.9(4)
Yb-O(1) ^{#7}	2.478(8)	O(4)-Yb-O(1) ^{#4}	76.0(3)		
		O(4) ^{#3} -Yb-O(1) ^{#4}	84.5(3)		
		O(3) ^{#1} -Yb-O(1) ^{#5}	72.7(3)		
		O(3) ^{#2} -Yb-O(1) ^{#5}	130.7(3)		
		O(4)-Yb-O(1) ^{#5}	84.5(3)	O(2) ^{#8} -K-O(2) ^{#9}	109.4(4)
		O(4) ^{#3} -Yb-O(1) ^{#5}	76.0(3)	O(2) ^{#8} -K-O(2) ^{#6}	77.1(2)
Mo-O(2)	1.722(9)	O(1) ^{#4} -Yb-O(1) ^{#5}	154.1(4)	O(2) ^{#9} -K-O(2) ^{#6}	121.64(13)
Mo-O(4)	1.766(9)	O(3) ^{#1} -Yb-O(1) ^{#6}	84.2(3)	O(2) ^{#8} -K-O(2) ^{#7}	121.64(13)
Mo-O(3)	1.785(8)	O(3) ^{#2} -Yb-O(1) ^{#6}	69.6(3)	O(2) ^{#9} -K-O(2) ^{#7}	77.1(2)
Mo-O(1)	1.848(8)	O(4)-Yb-O(1) ^{#6}	71.2(3)	O(2) ^{#6} -K-O(2) ^{#7}	149.7(4)
		O(4) ^{#3} -Yb-O(1) ^{#6}	138.8(3)	O(2) ^{#8} -K-O(3) ^{#2}	136.4(3)
		O(1) ^{#4} -Yb-O(1) ^{#6}	117.3(3)	O(2) ^{#9} -K-O(3) ^{#2}	106.6(3)
		O(1) ^{#5} -Yb-O(1) ^{#6}	70.7(3)	O(2) ^{#6} -K-O(3) ^{#2}	63.3(3)
		O(3) ^{#1} -Yb-O(1) ^{#7}	69.6(3)	O(2) ^{#7} -K-O(3) ^{#2}	89.7(3)
K-O(2) ^{#8}	2.679(10)	O(3) ^{#2} -Yb-O(1) ^{#7}	84.2(3)	O(2) ^{#8} -K-O(3) ^{#1}	106.6(3)
K-O(2) ^{#9}	2.679(10)	O(4)-Yb-O(1) ^{#7}	138.8(3)	O(2) ^{#9} -K-O(3) ^{#1}	136.4(3)
K-O(2) ^{#6}	2.790(10)	O(4) ^{#3} -Yb-O(1) ^{#7}	71.2(3)	O(2) ^{#6} -K-O(3) ^{#1}	89.7(3)
K-O(2) ^{#7}	2.790(10)	O(1) ^{#4} -Yb-O(1) ^{#7}	70.7(3)	O(2) ^{#7} -K-O(3) ^{#1}	63.3(3)
K-O(3) ^{#2}	2.825(10)	O(1) ^{#5} -Yb-O(1) ^{#7}	117.3(3)	O(3) ^{#2} -K-O(3) ^{#1}	58.3(3)
K-O(3) ^{#1}	2.825(10)	O(1) ^{#6} -Yb-O(1) ^{#7}	146.9(4)		

Symmetry transformations to generate equivalent atoms: ^{#1} -x,-y,-z; ^{#2} x,-y,z+1/2; ^{#3} -x,y,-z+1/2; ^{#4} -x+1,y,-z+1/2; ^{#5} x-1,y,z; ^{#6} -x+1,-y,-z; ^{#7} x-1,-y,z+1/2; ^{#8} -x+1/2,y-1/2,z; ^{#9} x-1/2,y-1/2,-z+1/2

3.4. Vibrational properties

The free $[\text{MoO}_4]^{2-}$ ion with Td point symmetry is characterized by four internal fundamental vibration modes $\nu_1(A_1)$, $\nu_2(E)$, $\nu_3(F_2)$, and $\nu_4(F_2)$, one free rotation mode and one translation mode (F_2) [35, 36]. As the MoO_4 units are slightly distorted tetrahedra in the crystal structure of $\text{KYb}(\text{MoO}_4)_2$, their local symmetry is lowered. It thus results in the splitting of all the degenerate vibrations due to the crystal field (CF) effect [37, 38].

Intense bands are visible in the Raman spectrum of $\text{KYb}(\text{MoO}_4)_2$ registered under ambient conditions for a selected single crystal, Figure 7. This spectrum presents a strong similarity with that of $\text{KY}(\text{MoO}_4)_2$, in agreement with the isostructural nature of these two orthorhombic compounds with a "polymeric nature" due to the presence of $[\text{LnO}_8]_n$ chains [33, 39]. It should be mentioned that no Raman band is observed for $\text{KYb}(\text{MoO}_4)_2$ in the frequency range over 1000 cm^{-1} . The maximal phonon frequency of 946 cm^{-1} corresponds to rather low energy, indicating that $\text{KYb}(\text{MoO}_4)_2$ material could be a suitable host for good upconverting luminescence properties [40].

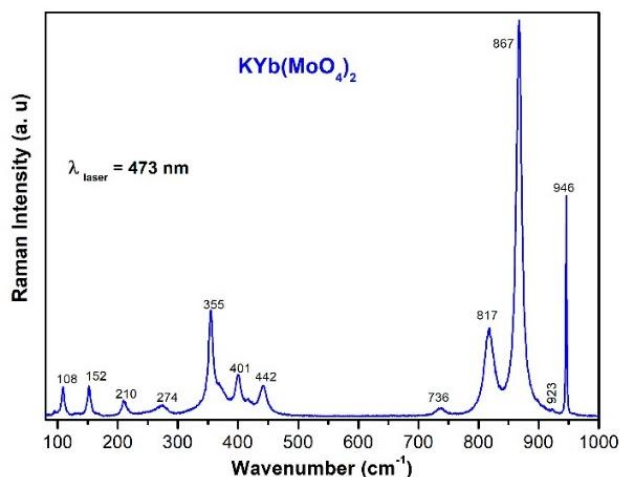


Figure 7. Unpolarized Raman spectrum registered at room temperature.

According to previous papers on vibrational studies of molybdates, the bands above 700 cm^{-1} are assigned to internal symmetric and antisymmetric stretching modes of the O atoms of the tetrahedral molybdate anions $(\text{MoO}_4)^{2-}$ [33, 39]. Internal symmetric and asymmetric O–Mo–

O bending vibration modes of the $(\text{MoO}_4)^{2-}$ units are observed in the 300 - 450 cm^{-1} frequency region [33]. The weak band observed at 736 cm^{-1} could be assigned to the stretching vibrations of the double-bridge in the "Mo₂O₈ dimers" (Figure 5) as already reported for the yttrium double molybdate [33, 34]. Finally, the bands below 300 cm^{-1} may originate from translational motions of K⁺, Yb³⁺, $(\text{MoO}_4)^{2-}$ ions, and lattice rotational modes as already reported [33].

3.5. Magnetic properties

The magnetic properties of the double molybdate KYb(MoO₄)₂ have been characterized by the determination of its magnetic susceptibility. The magnetization was measured for a sample placed in a constant magnetic field H of 0.5 T at temperatures ranging from 1.8 to 300 K. The magnetization depends on the size and the respective alignment of the dipolar atomic moments. The electronic configuration of the paramagnetic Yb³⁺ isolated ion is [Xe] 4f¹³ with [Xe] = 2s²2p⁶3s²3p⁶3d¹⁰4s²4p⁶4d¹⁰, which yields the spectral term ²F_{7/2} as the ground state, and the effective magnetic moment μ_{eff} of Yb³⁺ is equal to 4.54 μ_{B} .

No variation was observed between the data collected after cooling the sample in a magnetic field (FC) and the data collected after cooling in a zero field (ZFC). The magnetic susceptibility was corrected from the diamagnetic contribution of KYb(MoO₄)₂ estimated to be $\chi_{\text{dia}} = -2.66 \cdot 10^{-4}$ emu/mol.

Figure 8(a) shows that the Yb molar dc magnetic susceptibility χ of a 381 mg sample of as-grown KYb(MoO₄)₂ crystals, measured in a constant magnetic field H of 5000 Oe, is temperature-dependent. On cooling up to 40 K, the susceptibility curve shows a continuous and slow increase. In the low-temperature range 1.8 - 40 K, the magnetic susceptibility increases sharply.

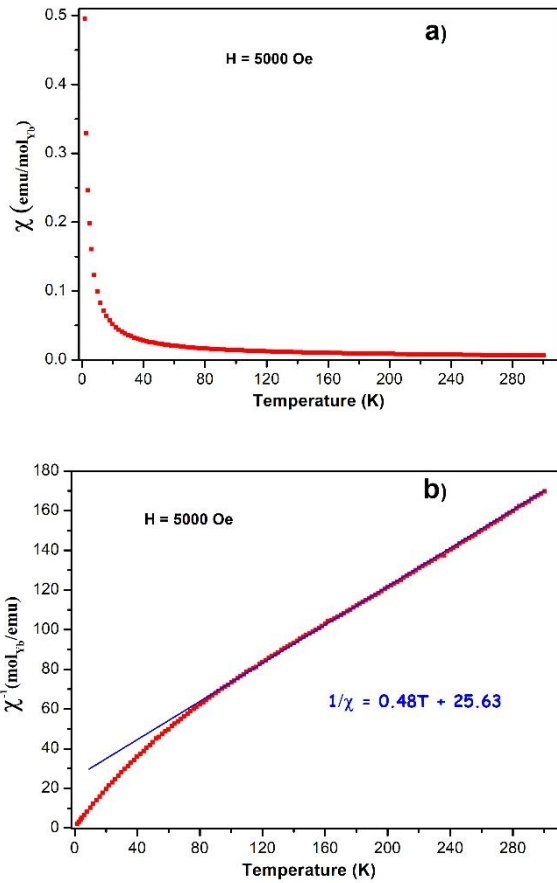


Figure 8. Magnetic (a) and inverse (b) susceptibility χ of $\text{KYb}(\text{MoO}_4)_2$ in a constant H field as a function of the temperature, with a Curie-Weiss fit (blue line) of the high-temperature data.

For the high-temperature range, we notice a linear behavior of the inverse magnetic susceptibility χ^{-1} from 300 K down to 80 K, Figure 8(b). The data obey a Curie-Weiss law $\chi_{\text{mol}} = C/(T-\theta_p)$ with the least-squares parameters $C = 2.08(1) \text{ emu}\cdot\text{mol}^{-1}$ (Curie constant) and $\theta_p = -52 \text{ K}$ (paramagnetic Curie-Weiss temperature). The experimental effective magnetic moment μ_{eff} of $4.1(1) \mu_B/\text{Yb}$, calculated for $\text{KYb}(\text{MoO}_4)_2$ from the Curie constant ($\mu_{\text{eff}} = (8 \times C)^{1/2}$), is in good agreement with the μ_{eff} value expected for the Yb^{3+} free-ion. This result underlines the weak effect, in the 80-300 K domain, of the crystal field (CF) environment on the magnetic properties of the f electrons in the Yb-based double molybdate under study.

A very large negative value of θ_p has already been reported for Yb-based oxides [40, 41]. Such "exceptional compartment" of Yb^{3+} could suggest an antiferromagnetic coupling between the magnetic moments. In the orthorhombic structure of $\text{KYb}(\text{MoO}_4)_2$, the Yb sites are separated by $3.9358(2) \text{ \AA}$, a rather short distance for Yb...Yb interactions, which could lead to an antiferromagnetic coupling between the Yb^{3+} cations, Figure 5. However, this antiferromagnetic coupling should be weak, and thus, the large negative θ_p value may have other origins. Alternative explanations could be thought in term of isolated Yb^{3+} centers in the lattice as already described for Sr_2YbMO_6 ($M = \text{Nb, Ta, Sb}$) materials [40], and of the crystalline electric field effect on the $^2F_{7/2}$ ground state of Yb^{3+} ions as already reported [15, 40-43]. The arrangement of oxygen around the Yb^{3+} ions would give rise to a splitting of the $^2F_{7/2}$ ground state at low temperatures.

The field-dependent M - H curves plotted for $\text{KYb}(\text{MoO}_4)_2$ at different temperatures ranging from 1.8 K up to 300 K are given in Figure 9. Several magnetic regimes are observed. The magnetization M measured at $T = 1.8 \text{ K}$ displays a large field dependence with a saturation zone starting from $H = 40 \text{ kOe}$ up to the highest applied magnetic field $H = 50 \text{ kOe}$. The saturation reaches $1.8 \mu_B / \text{Yb}$ at 5T. At 1.8 K, the magnetic alignment becomes geometrically frustrated.

At $T = 300 \text{ K}$, the magnetization is a linear function of the field H within the whole magnetic field range investigated in accordance with a paramagnetic behavior.

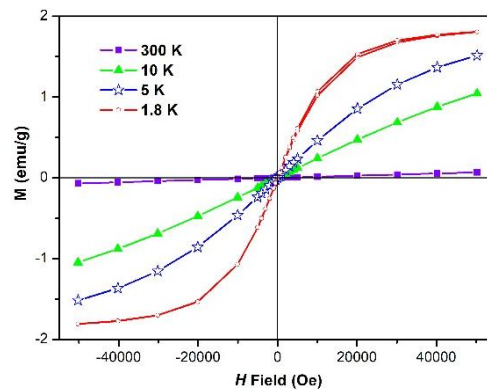


Figure 9. M-H curves recorded at different temperatures for crushed KYb(MoO₄)₂ crystals.

4. Conclusion

KYb(MoO₄)₂ double molybdate was obtained for the first time as transparent and colorless single crystals, using a high-temperature solution technique. The flux-grown material crystallizes in the orthorhombic symmetry with the Mo^{VI} valence state and displays an ordered distribution of the K⁺ and Yb³⁺ cations, both on C₂ symmetry positions. The structural type KY(MoO₄)₂ was assigned based on single-crystal structure determination. The existence of "Mo₂O₈ dimers" was confirmed by the presence of the 736 cm⁻¹ Raman band assigned to the stretching vibrations of Mo-O double-bridge interactions. Magnetic properties of KYb(MoO₄)₂ have been investigated from 1.8 to 300 K. The temperature-dependent magnetic susceptibility follows a Curie–Weiss law from 80 to 300 K, with a very large negative paramagnetic constant θ_p of -52 K. Such exceptional comportment of Yb³⁺ would suggest an antiferromagnetic spin-order coupling between the Yb³⁺ ions. A magnetic neutron diffraction study would help to clarify the existence of a potential ordered magnetic state in KYb(MoO₄)₂.

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